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## The design of $Zn_2SiO_4$ pigments for thermal control coatings

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### ABSTRACT

$Zn_2SiO_4$  pigments were synthesized from  $ZnO$  and  $SiO_2$  powders obtained from different manufacturers at various concentration ratios and differing in particle size and specific surface area. The optimal values of particle size and specific surface area were determined, thus making it possible to obtain the pigments with a small value of solar absorptance ( $\alpha_s$ ). The  $\alpha_s$  values of the synthesized pigments were compared with other types of pigments used to design thermal control coatings. The study showed that it is possible to synthesize  $Zn_2SiO_4$  powders with high reflectivity in a wide spectral range (200–2500 nm) and a small value of solar absorptance ( $\alpha_s = 0.077$ ) - significantly lower compared to  $ZnO$  powders - using a solid-state method from zinc oxide micropowders and silicon dioxide nanopowders.

### KEYWORDS

thermal control coatings • pigments • synthesis • diffuse reflectance spectra • solar absorptance

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## Introduction

The temperature of space objects is maintained at a given level by thermal control systems, which include active and passive components [1,2]. The active component is composed of cooling gases, pumps, gas pipelines, and check valves. It is activated in emergencies upon temperature rising [3]. As for the passive thermal control component, it plays the main role in maintaining the temperature at a certain level. It takes the form of a coating applied to thermal control radiators, housings, and individual devices. This coating operates throughout the entire period of orbital flights. Thermal control coatings (TCC) reflect the electromagnetic radiation of the Sun and re-emits the heat generated during the operation of the on-board equipment. The temperature of the objects is solely determined by the absorption of solar energy and by the re-emission of heat released by the on-board equipment. According to the Stefan–Boltzmann law, it can be determined via the following expression:  $T \sim k \cdot (\alpha_s / \varepsilon)^{1/4}$ , where  $k$  is the coefficient comprising the areas of absorbing and emitting surfaces,  $\alpha_s$  is the value of solar absorptance,  $\varepsilon$  is the emissivity of TCC.

If other conditions are kept equal, the temperature is considered proportional to  $\alpha_s$  absorption coefficient. Therefore, it is necessary to reduce the value of  $\alpha_s$  in order to decrease the area of both TCC and thermal control radiators and to maintain the required temperature. According to international standards [4,5], the values of absorptance should



fall within  $\alpha_s \leq 0.2$  for TCCs of the optical solar reflector (OSR) class. This requirement is met for the existing conventional TCCs. However, it is necessary to design the coatings with lower  $\alpha_s$  values in order to reduce their area, weight, and cost. The TCCs based on dielectric and semiconductor pigments with  $\alpha_s$  varying from 0.12 to 0.15 are currently used in thermal control systems [6–8]. To design TCCs with even lower absorption coefficient values, various combinations of components have been recently used in the pigment design. The usage of these components allows one to obtain  $\alpha_s \leq 0.1$ .

The second requirement for modern OSR TCCs is the high stability of absorption coefficient under operating conditions. A number of methods have been recently developed to increase the radiation stability of TCCs, their pigments and binders when they are exposed to solar spectrum quanta and charged particles. These methods are mainly based on three principles:

1. on the modification of TCCs by nanoparticles that act as small-size relaxation centers for primary photo- and radiation defects caused by irradiation;
2. on the modification of TCCs by rare earth elements (REE) that absorb the primary radiation-induced defects;
3. on the formation of protective layers with higher radiation stability on the surface of TCCs. If the composition of TCC includes nanoparticles or  $\text{SiO}_2$  particles, which form protective layers on the surface of various compounds [9–12], then it becomes possible to significantly increase the radiation stability of such compounds.

$\text{SiO}_2$  powders have a large band gap [13], therefore, they do not absorb the main part of the solar spectrum and are prone to form protective layers on the surface of inorganic compounds [14].  $\text{ZnO}$  powders with a certain particle size distribution show a high reflectance in solar spectrum and sufficient radiation stability when exposed to charged particles and solar spectrum quanta [15–19]. They can be used in the design of highly reflective pigments that can demonstrate stability under ionizing radiation. Therefore,  $\text{ZnO}$  and  $\text{SiO}_2$  powders might appear to be the most suitable components for designing pigments for OSR TCCs.

In view of the above, the purpose of this work was to develop a technology for the synthesis of pigments based on  $\text{ZnO}$  and  $\text{SiO}_2$  powders to be applied in TCCs with ultra-low  $\alpha_s$  values.

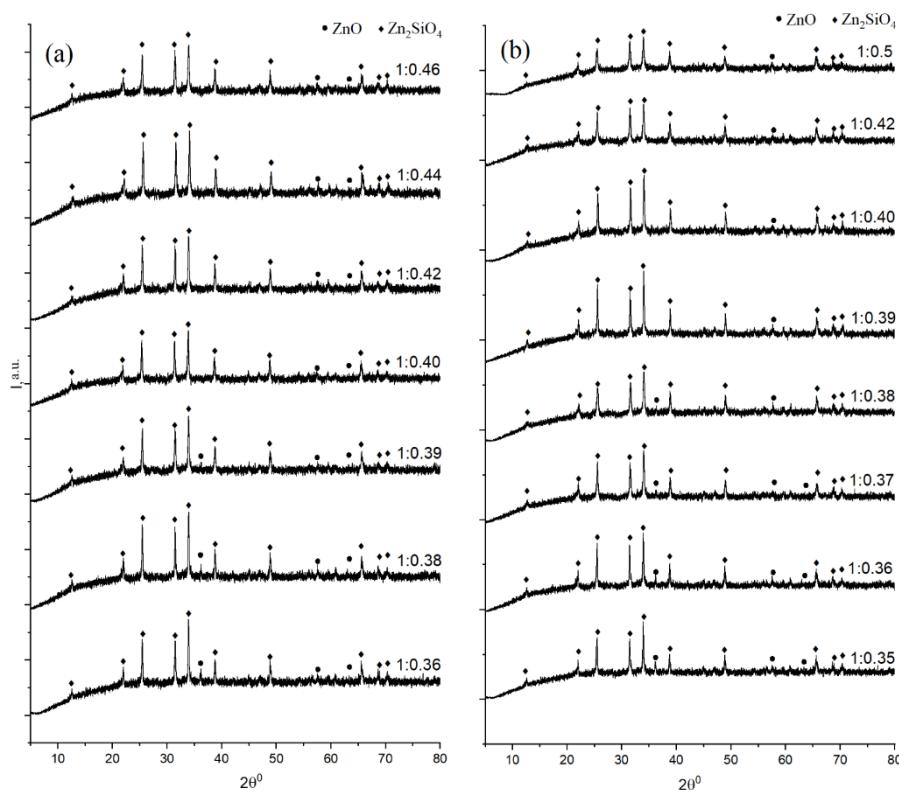
## Materials and Methods

$\text{Zn}_2\text{SiO}_4$  powders were synthesized by solid-state method from  $\text{ZnO}$  micropowder of PA grade (pure for analysis: average particle size 1.3  $\mu\text{m}$ ; weight percent of the active ingredient not less than 99.67 %) and silicon dioxide nanopowders of two brands: Aerosil A-300 (particle size of 5–20 nm, specific surface area of 300  $\text{m}^2/\text{g}$ ) and Plasmoetherm (particle size of 10–12 nm, specific surface area of 180–220  $\text{m}^2/\text{g}$ ). The ratio of zinc oxide and silica powders was selected to vary from 1:0.35 to 1:0.5. To obtain zinc orthosilicate,  $\text{ZnO}$  micropowder was mixed with  $\text{SiO}_2$  ( $n\text{SiO}_2$ ) nanopowder in a ceramic mortar for 20 min. Next, the resulting mixture was heated in a muffle furnace at a temperature of 1100 °C for 2 h. After cooling, the obtained  $\text{Zn}_2\text{SiO}_4$  powder was ground in a ceramic mortar for 10 min.

X-ray diffraction (XRD) spectra of the powders obtained from  $ZnO + nSiO_2$  components were recorded for all their concentration ratios using Shimadzu XRD-6100 X-ray diffractometer. The diffuse reflectance spectra ( $\rho_\lambda$ ) were recorded via Shimadzu UV-3600 Plus spectrophotometer with ISR-603 diffuse reflectance attachment within 0.2 to 2.5  $\mu\text{m}$ . Solar absorptance was calculated from  $\rho_\lambda$  spectra using international standards [4,5] and Johnson tables [20].

## Results and Discussion

The XRD spectra of  $Zn_2SiO_4$  micropowders (Fig. 1) showed the peaks assigned to willemite – a  $Zn_2SiO_4$  compound in its crystalline modification [21,22]. This modification was in trigonal syngony [23] with space group R-3 [24]. Additionally, the peaks were observed that could be assigned to  $ZnO$  crystalline structure in cubic phase [25].

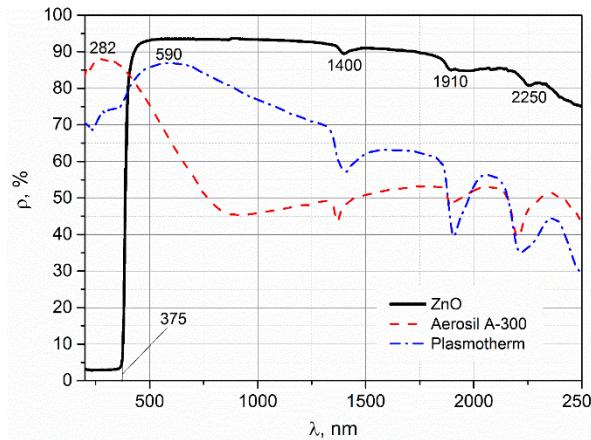


**Fig. 1.** XRD spectra of  $Zn_2SiO_4$  powders obtained at different concentrations of Plasmotherm (a) and Aerosil (b)  $SiO_2$  brands

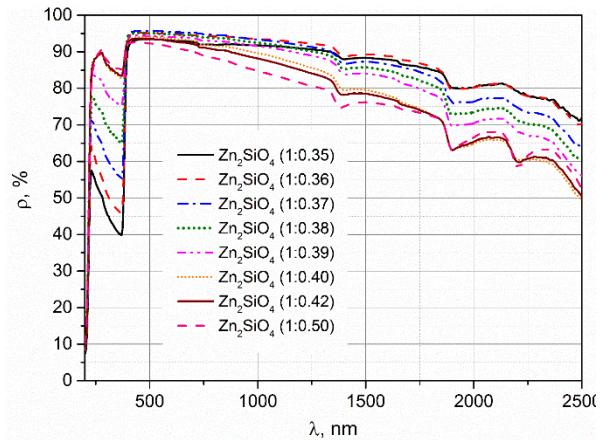
Figure 1 demonstrates that the intensity of peaks assigned to willemite increases whereas the intensity of peaks of cubic  $ZnO$  decreases with an increase in the concentration of  $SiO_2$  nanopowder (regardless of  $SiO_2$  nanopowder brand). Thus, increasing the amount of the initial  $SiO_2$  nanopowder during synthesis leads to an increase in the concentration of the target compound (willemite).

The diffuse reflection spectra of the powders used in the synthesis (Fig. 2) confirmed that the main absorption edge of the  $ZnO$  micropowder was 375 nm, which was consistent with the known values obtained by various authors [26–28]. In the longer wavelength

region, the reflectance increased and reached 93 % at  $\lambda \sim 500$  nm. Within  $\lambda \geq 1000$  nm, it gradually decreased and reached 75 % at  $\lambda = 2500$  nm. The bands appeared in the spectrum at 1400, 1740, and 2240 nm, due to the absorption of OH groups sorbed on the powder surface [29–32].



**Fig. 2.** The diffuse reflectance spectra for ZnO micropowder, Aerosil A-300 nanopowder, and Plasmoetherm nanopowder



**Fig. 3.** The diffuse reflectance spectra for  $\text{Zn}_2\text{SiO}_4$  powders obtained at different concentrations of Aerosil A-300  $\text{SiO}_2$  nanopowder

The  $\rho_\lambda$  spectra of  $\text{SiO}_2$  nanopowders of Aerosil A-300 and Plasmoetherm brands were characterized by a large band gap, therefore, their main absorption edge was within  $\lambda < 200$  nm. The highest value of reflectance for Aerosil A-300 powder was assigned to 282 nm and reached 88 %. It decreased with an increase in the wavelength and reached the lowest value (45 %) at  $\lambda = 900$  nm. In the longer wavelength region, the reflectance coefficient slightly increased, while the absorption bands of the OH groups were recorded at 1380, 1910 and 2220 nm similar to the reflectance spectrum of ZnO powder. However, the intensity of these bands for Aerosil A-300 powder was significantly higher.

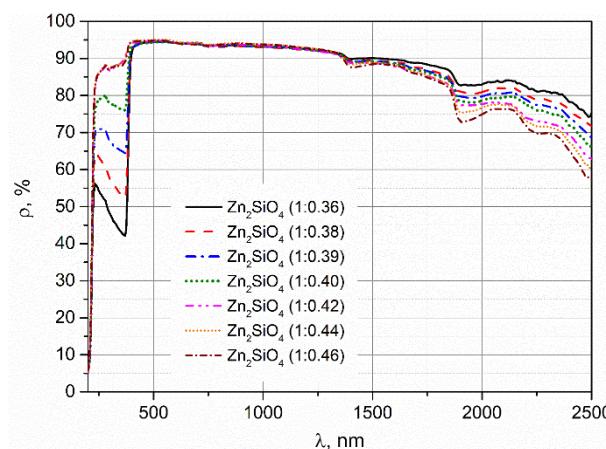
The  $\rho_\lambda$  spectrum for  $\text{SiO}_2$  nanopowder of Plasmoetherm brand was different from the one for Aerosil A-300 by the following features: the highest value of reflectance was assigned to 590 nm, that is, it was shifted to the long-wave region; the reflectance coefficient in the longer wavelength region decreased and reached only 30 % at  $\lambda = 2500$  nm. Identical absorption bands of OH groups were recorded in the reflectance spectrum of this powder, however, they displayed a much higher intensity.

The reflectance spectra of the obtained  $\text{Zn}_2\text{SiO}_4$  powders were characterized by the following features. The main absorption edge was clearly visible for ZnO powder at the same wavelength as for the initial powder spectrum (375 nm). The highest value of reflectance was achieved at a slightly lower wavelength (410–430 nm) in comparison with the initial ZnO powder (500 nm). The reflectance coefficient gradually decreased after the highest value and up to 2500 nm. Its lowest value was consistent with the concentration of  $\text{SiO}_2$  nanopowder during synthesis: the higher the powder concentration, the lower the reflectance coefficient.

Distinct patterns were traced in the shorter wavelength region from the main absorption edge of ZnO powder ( $\lambda = 375$  nm). These patterns were presumably caused by the contribution of  $\text{SiO}_2$  nanopowder to the reflectance spectrum of the obtained

compound  $Zn_2SiO_4$ . The reflectance increased with a decrease in the wavelength from the main absorption edge of the  $ZnO$  powder, reached the highest value, and then decreased. The highest value of reflectance increased with an increase in the concentration of  $SiO_2$  nanopowder, i.e., the reflection spectrum peaked in the short-wave region. For  $Zn_2SiO_4$  powder obtained using Aerosil A-300 nanopowder (Fig. 3), the absolute value of the reflectance was 57, 66, 71, 78, 83, 88, and 88% at the ratio of zinc oxide and silicon dioxide nanopowder concentrations of 1:0.35, 1:0.36, 1:0.37, 1:0.38, 1:0.39, 1:0.40, 1:0.42, 1:0.50, respectively.

The diffuse reflectance spectra of  $Zn_2SiO_4$  powders obtained at different concentrations of Plasmoetherm  $SiO_2$  nanopowder (Fig. 4) were qualitatively the same as those for the powders obtained at different concentrations of Aerosil A-300  $SiO_2$  nanopowder brand (Fig. 2).



**Fig. 4.** The diffuse reflectance spectra of  $Zn_2SiO_4$  powders obtained at different concentrations of Plasmoetherm  $SiO_2$  nanopowder

The absolute values of the reflectance coefficient in different regions of the spectrum were consistent with the concentration of  $SiO_2$  nanopowder during synthesis. They were slightly different from each other and from their respective values in the spectra of  $Zn_2SiO_4$  powders synthesized using Aerosil A-300 nanopowder. Therefore, it can be concluded that the synthesis of  $Zn_2SiO_4$  powders obtained on the basis of  $ZnO$  micropowder and  $SiO_2$  nanopowders of Plasmoetherm and Aerosil A-300 brands at different concentrations allowed the authors to obtain pigments with high reflectance in the solar spectrum, depending on both the type of  $SiO_2$  nanopowder and on its concentration. Because solar absorptance is the performance characteristic of TCCs,

**Table 1.** The correlation between  $\alpha_s$  of  $Zn_2SiO_4$  powders and the concentration of  $SiO_2$  nanopowder of Aerosil A-300 brand

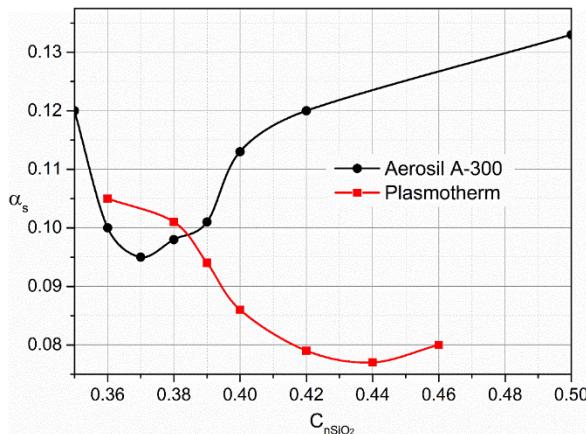
$ZnO:nSiO_2$	1:0.35	1:0.36	1:0.37	1:0.38	1:0.39	1:0.40	1:0.42	1:0.50
$\alpha_s$	0.120	0.100	0.095	0.098	0.101	0.113	0.120	0.133

**Table 2.** The correlation between  $\alpha_s$  of  $Zn_2SiO_4$  powders and the concentration of  $SiO_2$  nanopowder of Plasmoetherm brand

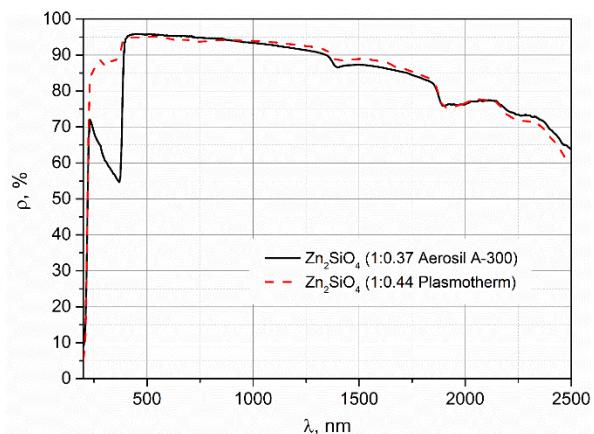
$ZnO:nSiO_2$	1:0.36	1:0.38	1:0.39	1:0.40	1:0.42	1:0.44	1:0.46
$\alpha_s$	0.105	0.101	0.094	0.086	0.079	0.077	0.080

it was calculated in relation to the type and concentration of  $\text{SiO}_2$  nanopowders used during synthesis. Tables 1 and 2 contain the results of the calculations performed.

Tables 1 and 2 show that an increase in the concentration of  $\text{SiO}_2$  nanopowders led to an increase in the absorption coefficient of the two powders obtained. The increase occurred with the lowest  $\alpha_s$  values of 0.095 at 1:0.37 for Aerosil A-300 powder and 0.077 at 1:0.44 for Plasmotherm powder. The correlation between  $\alpha_s$  and the concentration of  $\text{SiO}_2$  nanopowders (Fig. 5) confirmed that the use of Plasmotherm nanopowder made it possible to synthesize  $\text{Zn}_2\text{SiO}_4$  powders with significantly lower  $\alpha_s$  values.



**Fig. 5.** The correlation between the solar absorptance ( $\alpha_s$ ) of  $\text{Zn}_2\text{SiO}_4$  powders obtained and the concentration and type of  $n\text{SiO}_2$  powders used



**Fig. 6.** Diffuse reflectance spectra of  $\text{Zn}_2\text{SiO}_4$  powders with the lowest  $\alpha_s$  values obtained via different types of  $n\text{SiO}_2$  powder

This difference in  $\alpha_s$  values of  $\text{Zn}_2\text{SiO}_4$  powders was determined by the short-wave region of the diffuse reflectance spectra (Fig. 6). In this region (located at the main absorption edge of  $\text{ZnO}$  within 375 nm up to 200 nm), the reflectance coefficient of  $\text{Zn}_2\text{SiO}_4$  powders obtained using Plasmotherm nanopowder was higher compared to the ones obtained using Aerosil nanopowder.

## Conclusion

The hypothesis of the present work was that zinc oxide and silica powders could serve as the ideal starting materials for the synthesis of novel pigments to be used in thermal control coatings. Zinc oxide powders have been used as pigments for optical solar reflectors since 1960s both in Russia and in the United States who were managing their own full-scale space programs back then. These powders manifested sufficient radiation stability. Their disadvantage, however, was the large value of absorptance in the initial state ( $\alpha_s \sim 0.2$ ) determined by the semiconductor properties and the main absorption edge. In the short-wave region from the main absorption edge ( $\lambda \leq 375$  nm),  $\text{ZnO}$  powders absorbed 94.7 % of the energy from the radiation power of the entire solar spectrum, which is equal to  $0.139 \text{ W} \cdot \text{cm}^{-2}$  [33].

If synthesis is carried out using a  $\text{ZnO}$  powder with a dielectric powder with a large band gap (e.g.,  $\text{SiO}_2$  powder), one can obtain a compound that would show a significantly lower absorptance of solar spectrum in the UV region. This work has shown that it is possible to synthesize  $\text{Zn}_2\text{SiO}_4$  powders with high reflectance and a small value of the

solar absorptance using zinc oxide micropowders and silicon dioxide nanopowders via solid-state method. The solar absorptance of these powders would be almost 3 times less compared to  $ZnO$  powders ( $\alpha_s = 0.2$  and  $\alpha_s = 0.077$ , respectively).

Further research will be focused on the study of photo- and radiation stability of the powders obtained. The composition of  $Zn_2SiO_4$  powders allowed the authors to assume that they would demonstrate high radiation stability when exposed to ionizing radiation. The assumption can be grounded on the following factors. The powder contains a simple  $nSiO_2$  anion (i.e., not of complex composition). Such anion is unable to decompose into parts under irradiation. Therefore, no additional absorption centers might appear, as was established for complex anions of  $SO_4^{2-}$  type in  $BaSO_4$  [34,35] powders or  $CO_3^{2-}$  anions in  $CaCO_3$  [36] powders. If the concentration of  $nSiO_2$  allows for forming its monolayer or several layers the surface of grains and granules, then these layers might act as absorbing layers with high radiation stability [37]; as small-size defects being the relaxation centers for the primary photo- and radiation defects [38].

## CRediT authorship contribution statement

**Mikhail M. Mikhailov**  : supervision, conceptualization, writing – original draft, writing – review & editing; **Alexey V. Filimonov**  : supervision, writing – review & editing; **Alexey N. Lapin**  : data curation, investigation, writing – original draft; **Semyon A. Yuryev**  : writing – review & editing, validation; **Vladimir A. Goronchko**  : writing – review & editing, visualization; **Dmitriy S. Phedosov**: investigation, visualization.

## Conflict of interest

The authors declare that they have no conflict of interest.

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